

Raman Spectra of $\text{Li}_2\text{O-SiO}_2$ Glasses Prepared by Rapid Quenching

Masahiro TATSUMISAGO,* Tsutomu MINAMI,

Norimasa UMESAKI,† and Nobuya IWAMOTO†

Department of Applied Chemistry, University of
Osaka Prefecture, Mozu-Umemachi, Sakai, Osaka 591†Welding Research Institute, Osaka University,
Mihogaoka, Ibaraki, Osaka 567

Raman spectra were measured of rapidly quenched $\text{Li}_2\text{O-SiO}_2$ glasses containing high Li_2O concentrations. The proportions of SiO_4 tetrahedral units with different numbers of non-bridging oxygen atoms per a silicon atom were determined for these glasses. It was found that silicates in the glass with the highest Li_2O content (63 mol% Li_2O) consist of large amounts of isolated SiO_4^{4-} and $\text{Si}_2\text{O}_7^{6-}$ ions.

Rapid quenching is one of the useful techniques to develop new glassy materials and to extend the composition range of glass formation. In the system $\text{Li}_2\text{O-SiO}_2$, we showed that the composition limit of glass formation is extended up to 66.7 mol% Li_2O ,^{2,3)} compared with 40 mol% Li_2O in the usual quenching method. These new glasses obtained as a result of the extension of the glass-forming region are attractive with respect to their glass structure since the "network" of glass former is predicted to be little formed in such a composition.

This paper reports the Raman spectra of rapidly quenched $\text{Li}_2\text{O-SiO}_2$ glasses in the composition range 41-63 mol% Li_2O (analyzed values).

Glasses were prepared with a rapid-quenching apparatus combining a thermal-image furnace and a twin roller, the details of which were reported elsewhere.²⁾ The lithium contents of the obtained glasses were determined by atomic absorption analysis. Raman spectra were measured with a JASCO model R-800 double-grating spectrophotometer at a scattering angle of 90° . The excitation source was 5145 Å

line of an NEC GLG-3300 mode Ar^+ ion laser at power levels from 300 to 400 mW. Each spectrum obtained was deconvoluted into several Gaussian peaks based on the least-squares procedure.

Figure 1 shows the Raman spectra of the $\text{Li}_2\text{O}-\text{SiO}_2$ glasses prepared by rapid quenching. In each spectrum several scattering peaks are observed at the two wavenumber regions, 500-750 and 800-1200 cm^{-1} . The peak at 500-750 cm^{-1} , assigned to the bridging Si-O-Si vibration⁴⁾ and labeled as B in Fig.1, becomes weaker and shifts to higher wavenumbers with an increase in the Li_2O content. This means that the content of bridging oxygens present in the glasses decreases with an increase in the Li_2O content. In the region 800-1200 cm^{-1} , four peaks seem to overlap. These peaks are assigned to the stretching mode of the non-bridging Si-O vibration.⁵⁾ It was shown that these peaks are attributable to the SiO_4 tetrahedral units with different numbers of non-bridging oxygen atoms (NBO) per a silicon atom in the units.^{6,7)} The four peaks seen in the spectra at 1030-1100, 950-970, 900-930, and 850-890 cm^{-1} are associated with the SiO_4 units with 1,2,3, and 4 NBO's, respectively.⁷⁾ The numerals given to the peaks in Fig.1 indicate NBO numbers in the units illustrated in the upper part of the figure. In glasses containing relatively small amounts of Li_2O , the main peaks are the absorption bands numbered as 1 and 2.

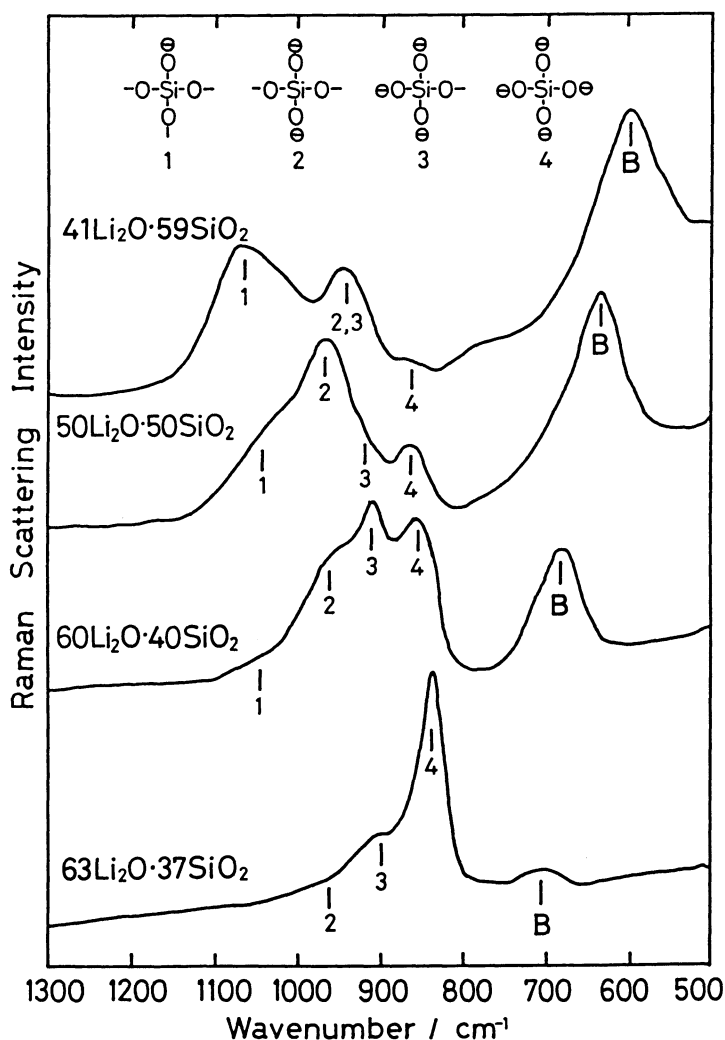


Fig. 1. Raman spectra of rapidly quenched $\text{Li}_2\text{O}-\text{SiO}_2$ glasses.

The spectra of these glasses are similar in shape to those of the $\text{Na}_2\text{O}-\text{SiO}_2$ glasses with the corresponding composition.^{8,9)} In glasses containing large amounts of Li_2O , on the other hand, the peaks 3 and 4 are mainly observed. These results indicate that the SiO_4 units with higher NBO/Si ratios increase with an increase in the Li_2O content. The glass with the highest Li_2O content, $63\text{Li}_2\text{O}\cdot 37\text{SiO}_2$, is considered to consist of isolated silicate ions such as SiO_4^{4-} (NBO/Si=4) and $\text{Si}_2\text{O}_7^{6-}$ (NBO/Si=3) and Li^+ ions. These "isolated ion" structures may be related to the fact that the T_g/T_1 ratios of the glasses largely deviate from the so-called "two-thirds rule."³⁾

Each spectrum was separated into four Gaussian peaks attributed to SiO_4 units with different NBO numbers and the proportions of these units were determined, assuming that the area of each peak is proportional to the amount of corresponding SiO_4 unit.^{5,10)} Figure 2

shows the proportions of the four SiO_4 units as a function of the Li_2O content. The contents of the structure units numbered as 2 (NBO/Si=2) and 3 (NBO/Si=3) have maxima at the compositions 50 and 60 mol% Li_2O , respectively. These compositions correspond to Li_2SiO_3 (NBO/Si=2) and $\text{Li}_6\text{Si}_2\text{O}_7$ (NBO/Si=3), respectively. It is concluded that a structure unit maximizes at the composition in which the ratio NBO/Si is identical with that of the structure unit.

This figure also suggests that any glass is not composed of only one structure unit.

For example, the glass $60\text{Li}_2\text{O}\cdot 40\text{SiO}_2$ is constructed not only

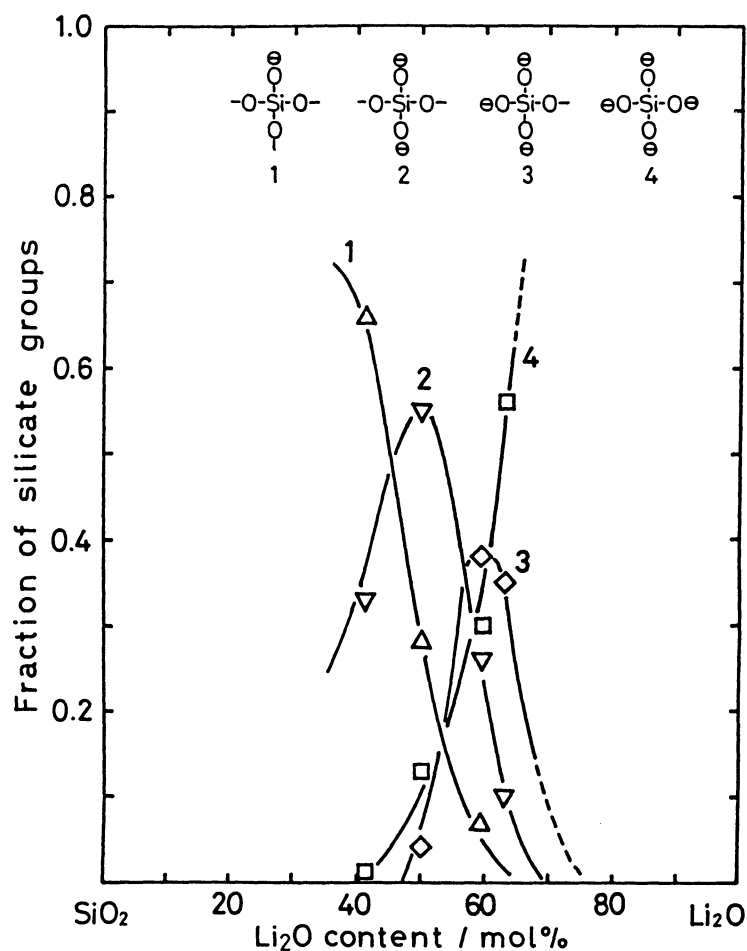
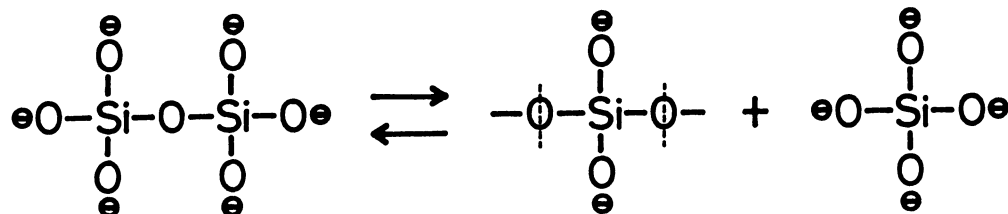


Fig. 2. Proportion of structure units present in $\text{Li}_2\text{O}-\text{SiO}_2$ glasses.

by the structure unit 3 (NBO/Si=3) but also by significant amounts of the structure units 2 (NBO/Si=2) and 4 (NBO/Si=4). Such a glass structure is considered to arise from an equilibrium such as



This kind of equilibrium would be present in the melts to be quenched. A similar situation was also shown in $\text{Li}_2\text{O}-\text{B}_2\text{O}_3$ glasses.¹⁰⁾

The authors acknowledge support to this research by a Grant-in-Aid for Scientific Research from the Ministry of Education, Culture and Science of Japan, and also by Nippon Sheet Glass Foundation for Materials Science.

References

- 1) R. T. Sarjeant and R. Roy, J. Am. Ceram. Soc., 50, 500 (1967).
- 2) M. Tatsumisago, T. Minami, and M. Tanaka, J. Am. Ceram. Soc., 64, C-97 (1981).
- 3) M. Tatsumisago, T. Minami, and M. Tanaka, Yogyo Kyokai Shi, 93, 581 (1985).
- 4) R. J. Bell and P. Dean, Discuss. Faraday Soc., 50, 55 (1970).
- 5) B. O. Mysen, D. Virgo, and C. M. Scarfe, Am. Miner., 65, 690 (1980).
- 6) N. Iwamoto, N. Umesaki, and K. Dohi, Yogyo Kyokai Shi, 92, 201 (1984).
- 7) Y. Tsunawaki, N. Iwamoto, T. Hattori, and A. Mitsuishi, J. Non-Cryst. Solids, 44, 369 (1981).
- 8) T. Furukawa, K. E. Fox, and W. B. White, J. Chem. Phys., 75, 3226 (1981).
- 9) S. A. Brawer and W. B. White, J. Chem. Phys., 63, 2421 (1975).
- 10) M. Tatsumisago, M. Takahashi, T. Minami, M. Tanaka, N. Umesaki, and N. Iwamoto, Yogyo Kyokai Shi, 94, 464 (1986).

(Received May 6, 1986)